Resumo
Materiais sustentáveis estão sendo investigados para substituir os materiais sintéticos, os compósitos com matriz polimérica têm sido aplicados em componentes, tais como capacetes e blindagem para os quais a resistência em um requisito importante. O presente trabalho avalia a classificação energética de impacto Charpy e Izod da matriz de epóxi compósitos DGEBA / TETA, reforçado com até 30% em volume de fibras de malva contínuas e alinhadas (Urena Lobata, L). Os testes de impacto foram realizados em amostras padronizadas em matriz de epoxi obtidas por aplicação de pressão em molde com cura a temperatura ambiente por 24 horas. Os resultados mostraram aumento significativo da energia de impacto com a fração de fibras incorporadas de malva. A superfície de fratura foi analisada por microscopia eletrônica de varredura, SEM. O desempenho das amostras incorporados com 30% de fibras, sofreu apenas ruptura parcial devido a dificuldades impostas ao quebrar-se as fibras, e, consequentemente, a natureza dos quais fendas tendem a propagar na interface entre a matriz de fibra / resina epóxi, o que ajuda a absorver a energia do impacto.

Palavras-chave: Malva; Compósitos; Matriz epóxi; Impacto Charpy; Impacto Izod; Ductilidade.

IMPACT ENERGY RATING CHARPY AND IZOD REACHED IN POLYMER COMPOSITES INCORPORATED WITH MALVA FIBERS

Abstract
Environmentally friendly materials are currently being investigated to replace synthetic materials, the polymer matrix composites have been applied in components such as helmets and shielding for which toughness in a major requirement. The present work evaluates the Charpy and Izod impact energy rating of epoxy matrix composites DGEBA/TETA, reinforced with up to 30% in volume of continuous and aligned malva fibers(Urena Lobata,L).The impact tests were performed on standard specimens obtained by pressing mold cure epoxy after 24 hours. The results showed significant increase in impact energy with the fraction of incorporated of malva fibers. The fracture surface was analyzed by scanning electron microscopy, SEM. The performance of the samples incorporated with 30% of fibers, suffered only partial rupture due to difficulty imposed by breaking the fibers themselves, and accordingly, the nature of which cracks tend to propagate at the interface between fiber/epoxy matrix, which helps to absorb impact energy.

Keywords: Malva; Composite; Epoxy matrix; Charpy impact; Izod impact; Notch toughness.

1 Doctor In Materials Engineering, Associate Professor, Department of Mechanical Engineering, ISECENSA, Campos dos Goytacazes, Rio de Janeiro, Brasil.
2 Graduating In Metallurgical Engineering and Materials, Student, Department of Materials, State University of the Northern Rio de Janeiro, Campos dos Goytacazes, Rio de Janeiro, Brasil.
3 Graduated In Chemical Engineering, Department of Chemical Engineering, Federal University of the Rio de Janeiro, Rio de Janeiro, Brazil.
4 Doctor In Materials Engineering, Associate Professor, Department of Electrical Engineering, Faculdade Redentor, Itaperuna, Rio de Janeiro, Brasil.
5 Phd In Materials Engineering, Emeritus Professor, Depart. of Materials, IME, Rio de Janeiro, RJ, Brazil.
1 INTRODUCTION

The modern industrial materials require more sustainable and environmentally friendly alternatives, this trend has increased the attention on reinforced polymer composites with lignocellulosic fibers, such as the malva Fibers (Urena Lobata, L). Those fibers composites present characteristics like a high impact absorption by the composite structure. This work aim to analyze the resistance to impacts of composites produced with epoxy matrix and amounts ranging from 0% to 30% in volume of continuous and aligned malva fibers. The objective is to analyze how the Charpy and Izod impact resistance improves with the inclusion of more and more fibers, up to 30% in volume. The results showed a remarkable increase in the notch toughness with the amount of incorporated malva fibers. This can be attributed to a preferential debonding of the fiber/matrix interface, which contributes to an elevated absorbed energy. Polymer composites reinforced with natural fibers, mainly those lignocellulosic obtained from plants, have been subject to extensive research works. Environmental, economical, societal and technical advantages of these composites are motivating their substitution for similar polymer composites reinforced with synthetic fibers. In particular, R&D efforts have been conducted aiming to replace glass fibers composites that are comparatively more abrasive to equipments, non-recyclable, more expensive, heavier and toxic. Among the many lignocellulosic fiber composites being investigated, those fabricated with malva fiber, ramie, coco, curaua, bamboo and others fibers longitudinally cut from the culm, still need information to support possible industrial applications. Several works, however, have been dedicated to those fibers as possible reinforcement of polymer composites. The figure 1 show a tipical plantation of Malva in the Amazonas State and Pará.

![Figure 1. A plantation of Malva fibers](image)

In spite of existing works on the properties of those fibers composites separated, the comparison of impact energy, between malva fibers reinforced in polymeric composites has yet to be evaluated. Therefore, the objective the present work was access energy of impact between charpy test and Izod with epoxy composites reinforced with different amounts malva fibers.

2 EXPERIMENTAL PROCEDURE

The malva fiber was obtained from Companhia Castanhal Sul, a firm that commercializes natural fibers cultivated in the north region of Brazil.
In order to produce composites with properties that are acceptable for industrial applications, the malva fibers were randomly selected to serve as reinforcement to an epoxy matrix. These fibers, with an average diameter of 0.065 mm, were laid down inside steel rectangular molds with distinct volume fractions were mixed in amounts of 0, 10, 20 and 30% in volume with epoxy resin type diglycidyl ether of the bisphenol–A (DGEBA) with a DER of 374 g/mol and an equivalent weight of 187.3 g/equiv. with triethylene tetramine (TETA), as hardener, in stoichiometric ratio corresponding to phr 13. Both DGEBA resin and TETA hardener were supplied by the DOW Chemical Co.

The fibers were maintained aligned along the length dimension in the dog bone shape composites. The still fluid mixture was poured onto the fibers inside a steel mold and allowed to cure at room temperature (RT) for 24 hours. Ten specimens were prepared for each volume fraction and each fiber kind. The fibers were continuously aligned along the length of the mold. The still fluid epoxy resin was poured onto the fibers and pressured in the mold up to 2 tons, creating the composites plates for each volume fraction. The already processed composites plates were allowed to undergo an initial cure at room temperature for 24 hours. Afterwards, a post-cure was conducted at 60°C for 4 hours. For each composite plate, 10 specimens were cut down, producing the specimen according to the ASTM D256 norm. The samples were impact tested in a PANTEC pendulum with Charpy and Izod configuration. The impact energy was obtained using an 15 J power hammer for composites with 0, 10, 20 and 30% of fibers. For each volume fraction of fibers, ten specimens were used for statistical validation. The figure 2 show the received and individually separated Malva fibers.

![Figure 2 (A) Malva fibers, and (B) Malva fibers individually separated.](image)

**3 RESULTS AND DISCUSSION**

Table I shows the results of the values of Charpy and Izod impact energy with their respective standard deviations for pure epoxy and composites with different volume fractions of malva fibers.
Table I – Charpy and Izod impact energy for epoxy composites reinforced with malva fibers.

<table>
<thead>
<tr>
<th>Fiber Percentage (%)</th>
<th>Charpy Impact Energy (J/m)</th>
<th>Izod Impact Energy (J/m)</th>
</tr>
</thead>
<tbody>
<tr>
<td>0</td>
<td>22.9 ± 9.7</td>
<td>21.8 ± 9.7</td>
</tr>
<tr>
<td>10</td>
<td>101.1 ± 28.7</td>
<td>60.2 ± 14.7</td>
</tr>
<tr>
<td>20</td>
<td>176.6 ± 41.1</td>
<td>111.9 ± 23.3</td>
</tr>
<tr>
<td>30</td>
<td>310.2 ± 98.1</td>
<td>169.4 ± 29.2</td>
</tr>
</tbody>
</table>

Based on the results of Table I, the Impact Energy variation as a function of the amount of the different fibers is shown in Figure 3.

Figure 3 – Charpy and Izod impact energy as a function of the amount of malva fibers.

One should notice the marked increase in Charpy impact energy with the fiber volume fraction of malva. It is also important to note that the error bars present the standard deviation, a common feature for lignocellulosic fibers. This is due to the heterogeneous nature of natural fibers, resulting in substantial dispersion properties of the composites reinforced by them.

In fact as shown in table 2, using long and aligned malva fibers for composites obtains relatively higher impact toughness composites with other fibers too.

Table 2 - Comparison on Impact toughness values for different fibers

<table>
<thead>
<tr>
<th>Composites 30% fibers</th>
<th>Impact Type</th>
<th>Impact Toughness (J/m)</th>
</tr>
</thead>
<tbody>
<tr>
<td>malva/epoxy</td>
<td>Charpy</td>
<td>310.2</td>
</tr>
<tr>
<td>malva/polyester</td>
<td>Charpy</td>
<td>716.2</td>
</tr>
<tr>
<td>ramie/epoxy</td>
<td>Charpy</td>
<td>211.7</td>
</tr>
<tr>
<td>ramie/polyester</td>
<td>Charpy</td>
<td>1004.8</td>
</tr>
<tr>
<td>coco/polyester</td>
<td>Charpy</td>
<td>241.2</td>
</tr>
<tr>
<td>coco/epoxy</td>
<td>Charpy</td>
<td>174.7</td>
</tr>
<tr>
<td>curaua/polyester</td>
<td>Charpy</td>
<td>169.7</td>
</tr>
<tr>
<td>curaua/epoxy</td>
<td>Charpy</td>
<td>103.2</td>
</tr>
</tbody>
</table>
It is important to discuss the macroscopic rupture characteristic of the specimens after the test. Figure 4 illustrates the characteristic of rupture of the epoxy specimens for each amount of fiber incorporated. In this figure is shown that the some specimens with 30% of malva fiber, i.e. the highest toughness obtained, do not separated into two parts after impact as showed in other work in literature 18. This indicates that cracks nucleated in the notch began to propagate across the brittle epoxy matrix, but when they reach the fiber interface, the crack changes direction.

Specimens with under 20% malva fiber incorporation undergo complete rupture. Specimens with 30% malva fibers incorporation, however, did not undergo complete rupture. This leads to the decrease in toughness observed in Figure 4. If all fibers were broken then the energy absorbed would have been even greater 19. The reason for having a crack nucleated at the notch, changing its trajectory to reach the fibers malva, and going to propagate through the interface with the matrix is due to the low interfacial resistance. This is a consequence of the incompatibility caused by the fact that lignocellulosic fibers are hydrophilic while the polymer matrix is hydrophobic 20.

Figure 4 shows the macrostructure appearance of the specimens after the Charpy and Izod impact tests for each volume fraction of fibers tested, it should be noticed that with volume fractions greater than 20% in test of fiber the fracture was not complete and the specimens did not brake, what shows that the impact energy was underestimated and should be higher if the specimens were completely separate.

The SEM analysis of the impact fracture permitted to have a better comprehension of the mechanism responsible for the higher toughness of epoxy composites reinforced with long and aligned malva fibers. Figure 5 shows the aspect of the fracture surface of a pure polyester (0% fiber) specimen21-24. With lower magnification, the lighter layer in the left side of the fractograph, Fig 5(A), corresponds to the specimen notch, revealing the machining parallel marks. The smoother and gray layer on the right side corresponds to the transversal fracture surface. The fracture in Fig. 5 suggests that a single crack was responsible for the rupture with the roughness in Fig 5(B), being associated with voids and imperfections during the processing.
Figure 5 – Fracture surface of the specimen pure epoxy (0% malva fiber): (A) general view with low increase (B) higher increase.

Figure 6 presents details of the impact fracture surface of a epoxy composite specimen with 30% of malva fiber. This fractograph shows an effective adhesion between the fibers and matrix, where cracks preferentially propagate. Some of the fibers were pulled out from the matrix and others were broken during the impact 25-26. By contrast, the part of the specimen in which the rupture preferentially occurred longitudinally through the fiber/matrix interface reveal that most of the fracture area is associated with the fiber surface. This behavior corroborates the rupture mechanism of cracks that propagate preferentially in between the malva fiber surface and the epoxy matrix due to the low interfacial strength 27. The greater fracture area, Fig.6, is associated with increasing amount aligned of malva fibers acting as reinforcement for the composite, that justify the higher impact energy absorbed. This behavior confirms the mechanism of rupture by cracks that due to the low interfacial shear stress, is preferably spread between the surface of the malva fibers and epoxy matrix as can be seen in Figure 6B 28.

This results in a longitudinal fracture area is relatively large compared to the transverse fracture of the specimens with up to 20% of malva fiber. Consequently have a higher impact energy to break an area comparatively higher as indicated by Yue ET AL(1995). Similar results were found in polyester matrix composites reinforced with malva fiber. This indicates that the malva fiber provide high tenacity to polymeric matrices reinforced by it.

Figure 6 – Fracture surface of the specimen epoxy composite (30% malva fiber): (A) general view with low increase (B) higher increase.
4 CONCLUSIONS

Composites made of continuous and aligned malva fibers as reinforcement in epoxy matrix, cured at room temperature, show a linear increase in notch toughness, measured in both Charpy and Izod impact tests in relation to the pure epoxy resin. The incorporation of malva fibers in epoxy matrix improves significantly the toughness of the composite compared to pure epoxy resin. Along with the retention of some whole malva fibers upon impact, low interfacial tension results in greater energy absorbed in due to the propagation of cracks in the fiber/matrix interface, allowing higher rupture area in relation to a transverse fracture that occurs in the matrix breaking. The incorporation of volume fractions exceeding 20% are associated with incomplete fracture of the specimens due to the flexibility of the malva fibers that despite the during the impact tests curving but are not ruptured.

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